

Title	Direct Observation of Ferromagnetic Spin Polarization in Gold Nanoparticles
Author(s)	Yamamoto, Y.; Miura, T.; Suzuki, M.; Kawamura, N.; Miyagawa, H.; Nakamura, T.; Kobayashi, K.; Teranishi, T.; Hori, H.
Citation	Physical Review Letters, 93(11): 116801-1-116801-4
Issue Date	2004-09-07
Type	Journal Article
Text version	publisher
URL	http://hdl.handle.net/10119/3988
Rights	Y. Yamamoto, T. Miura, M. Suzuki, N. Kawamura, H. Miyagawa, T. Nakamura, K. Kobayashi, T. Teranishi, and H. Hori, Physical Review Letters, 93(11), 116801, 2004. Copyright 2004 by the American Physical Society. http://link.aps.org/doi/10.1103/PhysRevLett.93.116801
Description	

Direct Observation of Ferromagnetic Spin Polarization in Gold Nanoparticles

Y. Yamamoto,¹ T. Miura,¹ M. Suzuki,² N. Kawamura,² H. Miyagawa,² T. Nakamura,² K. Kobayashi,²
T. Teranishi,^{1,*} and H. Hori¹

¹*School of Materials Science, Japan Advanced Institute of Science and Technology (JAIST), 1-1 Asahidai, Tatsunokuchi, Ishikawa, 923-1292, Japan*

²*Japan Synchrotron Radiation Research Institute (JASRI / SPring-8), 1-1-1 Kouto, Mikazuki, Sayou 679-5198, Japan*
(Received 24 March 2004; published 7 September 2004)

We report the first direct observation of ferromagnetic spin polarization of Au nanoparticles with a mean diameter of 1.9 nm using x-ray magnetic circular dichroism (XMCD). Owing to the element selectivity of XMCD, only the gold magnetization is explored. Magnetization of gold atoms as estimated by XMCD shows a good agreement with results obtained by conventional magnetometry. This evidences intrinsic spin polarization in nanosized gold.

DOI: 10.1103/PhysRevLett.93.116801

PACS numbers: 73.22.-f, 75.50.Tt, 75.70.Rf, 75.75.+a

Metal nanoparticles or clusters have recently gained great interest to physicists, chemists, and engineers with regards their practical applications, such as high density storage media [1] and next generation devices like single electron transistors [2], as well as their fundamental physics [3]. When the particle or cluster size is sufficiently small, degenerate energy levels split into discrete levels and the Fermi wavelength of the electrons becomes comparable to the system size. Consequently, the physical properties of these particles are expected to be quite different from the bulk properties. However, due to the difficulty of preparing samples with well-defined diameters, it has long been difficult to observe these differences experimentally. In the past two decades, a new class of materials, metal nanoclusters stabilized by chemical materials, have been extensively studied [4,5]. In particular, noble metal nanoclusters have been synthesized using various stabilizer molecules, *e.g.*, Au₅₅(PPh₃)₁₂Cl₆, Pd₅₆₁phen₃₆O_{200±20}, etc. [6]. Recently, a size adjustment method for noble metal (Pd, Pt, Au) nanoparticles embedded in linear alkyl chain polymers has been developed [7,8]. This method ensures the generation of monodispersed metal nanoparticles with arbitrary diameter. A wide range of particle diameters can be achieved and the variance in the diameter is very small. The ability of this method to adjust the cluster size allows a systematic investigation of the physical properties of metal clusters in the nanometer size range.

Hori *et al.* have reported the macroscopic magnetic properties of polymer-stabilized metal nanoparticles [9]. They observed superparamagnetic behavior in Pd and Au nanoparticles embedded in poly(*N*-vinyl-2-pyrrolidone) (PVP) at low temperatures. This means that the individual Pd or Au particles may have ferromagnetic moments, even though Pd shows paramagnetism and Au shows diamagnetism in the bulk state. However, there is no clear evidence that the ferromagnetism of Au particles is intrinsic, because their SQUID magnetization data may include signals originating from

trivial sources other than the nanoparticles, such as magnetic impurities or unpaired electrons in the protective polymer. The effects from these sources cannot be removed using conventional magnetometry techniques. An experimental technique that allows the selective observation of magnetization of Au nanoparticles should be used.

In this Letter, we present direct evidence of the intrinsic ferromagnetism of Au nanoparticles by means of element-specific magnetization (ESM) measurements based on the x-ray magnetic circular dichroism (XMCD) technique [10]. XMCD allows the detection of the magnetic moments of a particular element through sensitivity to the difference between the up- and down-spin densities around the Fermi level. This element selectivity is the most important advantage of this technique over conventional magnetometry and is essential in the present study for extracting the magnetization of the Au nanoparticles. Since the first observation of XMCD in Fe [11], the magnetism of nanostructured materials [12], thin films [13], and strongly correlated systems [14] have been widely investigated using this method.

The XMCD spectra were recorded using a highly sensitive spectrometer [15] installed at BL39XU of SPring-8 synchrotron radiation facilities. External magnetic fields up to 10 T were applied along the x-ray beam direction using a split-type superconducting magnet. The experimental resolution was high enough to detect XMCD signals of 10⁻⁵ parts of the spin-averaged x-ray absorption coefficients. This high sensitivity is achieved through the helicity modulation technique [15], based on lock-in detection and the high brilliance of the third generation synchrotron radiation source. We note that the modulation technique was crucial for the detection of extremely low ferromagnetism in Au nanoparticles as the technique dramatically improves the signal/noise ratio of the XMCD signal.

In the present work, Au nanoparticles protected by polyallyl amine hydrochloride (PAAHC) (abbreviated

as PAAHC-Au) were used as samples [Fig. 1(a)]. These were prepared by a method described in Ref. [7]. The distribution of particle diameter was determined from a transmission electron microscope image (TEM, Hitachi H-7100). The mean diameter and standard deviation of the particles were 1.9 nm (212 atoms per cluster) and 0.2 nm, respectively, measured by counting 200 particles from the TEM image [Fig. 1(b) and 1(c)]. The crystal structure of the nanoparticles was identified to be an *fcc* structure by x-ray powder diffraction (XRD, MAC Science) using Cu- $K\alpha$ radiation. Magnetization and susceptibility measurements were performed using a commercial superconducting quantum interference device (SQUID) magnetometer (Quantum Design, MPMS-XL) in the temperature range between 1.8 and 300 K and at a magnetic field of up to 7 T.

Figure 2 shows x-ray absorption spectroscopy (XAS) and XMCD spectra of Au at the L_{3-} ($2p_{3/2} \rightarrow 5d_{5/2}, 6s_{1/2}$ dipole allowed transitions) and L_{2-} edge ($2p_{1/2} \rightarrow 5d_{3/2}, 6s_{1/2}$) at 2.6 K in an applied magnetic field of 10 T. A negative XMCD signal was clearly observed at the L_{3-} edge (11.917 keV), whereas the XMCD signal at the L_{2-} edge (13.730 keV) was positive. The XMCD amplitude is of the order of 10^{-4} of the XAS step height.

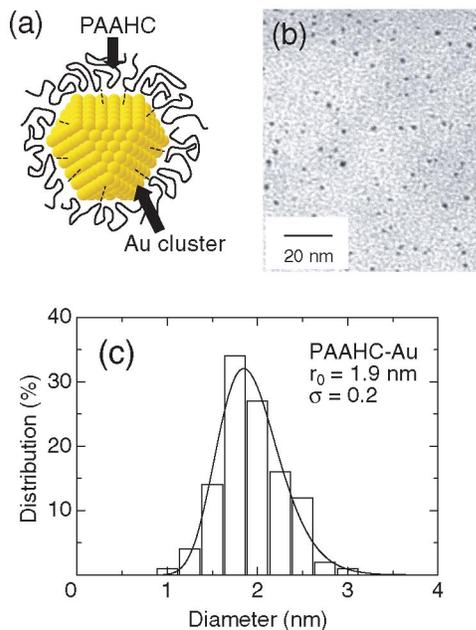


FIG. 1 (color online). (a) Schematic of PAAHC-Au nanoparticles. (b) TEM image of Au nanoparticles protected by PAAHC. Mean diameter of the sample is 1.9 nm. Scale bar is shown in the image. (c) Size distribution of PAAHC-Au nanoparticles. The solid line represents the fitting curve assuming the log-normal function $f(r) = \frac{1}{\sqrt{2\pi}r\sigma} \exp\left(-\frac{(\ln r - r_0)^2}{2\sigma^2}\right)$ where r is the particle diameter. The fitting results of parameters are the mean particle diameter $r_0 = 1.9$ nm and the standard deviation $\sigma = 0.2$ nm.

Small negative peak structures were observed at the high-energy side of the main peaks, at 11.930 and 13.741 keV. As shown by the dotted line in Fig. 2, the sign of the XMCD signal reversed when the magnetic field direction was changed. This verifies that the observed signal is truly of magnetic origin and does not arise from any artificial effects. To the best of our knowledge, this is the first evidence of an XMCD signal arising from Au.

The XMCD peak height at the L_{2-} edge is less than half that at the L_{3-} edge. This asymmetry indicates that the Au $5d$ electrons have considerable orbital moment. Using the sum rules [16,17], the ratio of the orbital to the spin magnetic moments was determined to be $\langle\mu_L\rangle/\langle\mu_S\rangle = 0.145$, which is larger than that in orbital-quenched $3d$ transition metals and is comparable to the reported values for Pd and Pt systems [18,19]. In determining $\langle\mu_L\rangle/\langle\mu_S\rangle$, the value of the hole number in the Au valence is not required and the magnetic dipole term $\langle T_z \rangle$ was neglected. In principle, the sum rules give separate quantitative values of the spin and orbital magnetic moments. However, this analysis requires an accurate value for the valence hole number and a close estimation of the $\langle T_z \rangle$ term. For Au nanoparticles, the hole number and $\langle T_z \rangle$ values are likely to be quite different from the known values for bulk Au [20], and hence we did not attempt to estimate these values. We will restrict further discussion to the relative variations of the XMCD signal as a function of external magnetic field or temperature.

Considering the XMCD signal to be proportional to the magnetization, ESM is obtained by recording the peak amplitude of the XMCD spectra at the Au L_{3-} edge as a function of external magnetic field and temperature.

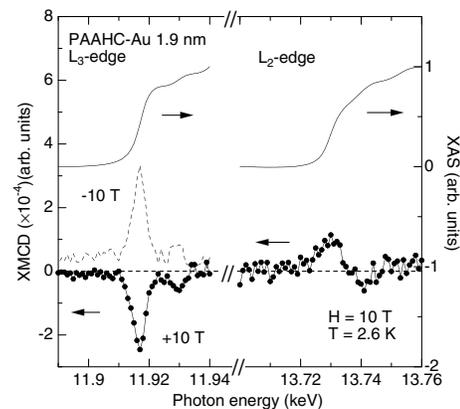


FIG. 2. XMCD and XAS spectra at the Au L_{3-} and L_{2-} edge with an applied magnetic field of 10 T. The XMCD and XAS spectra include a same scale factor so that the height of the edge jumps of the XAS spectra (solid lines) are unity. Solid lines with closed circle symbols represent an XMCD spectrum with an applied magnetic field of 10 T, while the dotted line indicates an XMCD spectrum in a magnetic field applied in the opposite direction, implying the observed signal is not an artificial effect.

Figure 3(a) shows ESM measurements scaled arbitrarily for comparison with the magnetic field variation of SQUID magnetization in the same figure. ESM increases with increasing magnetic field without saturation. This behavior is similar to the magnetization process obtained by SQUID magnetization measurements. There was no distinguishable hysteretic behavior in either the ESM or SQUID magnetization curve, while the XMCD spectra indicated a large magnetic anisotropy. We believe that this is because the superparamagnetic limit due to the small particle size obscured the magnetic anisotropy effect. As discussed in the following section, the magnetization process consists of superparamagnetic and temperature-independent Pauli-paramagnetic parts. For an ideal superparamagnetic system, the magnetization curve is expressed by $M(H) = N\mu L(x)$, with the Langevin function $L(x) = \coth(x) - 1/x$, where $x = \mu H/k_B T$, μ is the magnetic moment per particle and N is the total number of particles per unit mass. Hence, the total magnetization can be expressed as $M(H) = N\mu L(x) + \chi_{\text{Pauli}} H$. The dotted line in Fig. 3(a) represents the fitting results. The magnetic moment determined in this way was $0.4\mu_B/\text{Au}$ particle.

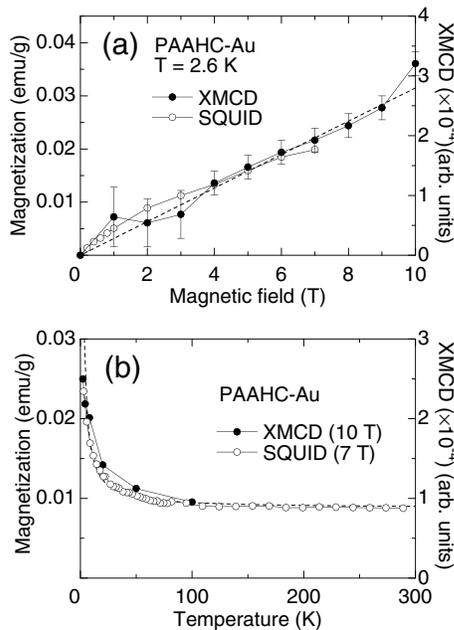


FIG. 3. (a) ESM of PAAHC-Au as a function of the applied magnetic field and magnetization process obtained by SQUID magnetometer. The integral of the peak intensity yields similar results. Dotted line is the fit to the data assuming a Langevin function plus a linear field-dependent term. (b) Temperature dependence of the XMCD peak intensity at 10 T recorded at the Au L_3 -edge and temperature variation of the magnetization measured by SQUID magnetometer at 7 T. Dotted line is the fit to the data assuming $\chi(T) = N\mu L(x)/H + \chi_{\text{Pauli}}$. Solid lines are guides for the eye.

XMCD is an element-selective technique and impurity atoms other than Au do not influence the experimental results. Since the value of the magnetic moment is relatively small, one might attribute the magnetization to the polarization of the conduction electrons of Au induced by magnetic impurities. In fact, XMCD has revealed induced spin polarization in nonmagnetic Cu layers in the case of the Co/Cu multilayer system [21]. However, this possibility can be excluded in the present system for the following reasons. First, the amount of magnetic impurity included in the sample was extremely small. We have determined the amount of magnetic impurities using an inductively coupled plasma mass spectrometer (ICP-MS), and Fe, Co, or Ni were not detected within the detectable limit of 5 ppm. Even if one assumes that the sample contains 5 ppm of Fe impurities, it is impossible to explain the absolute value of the magnetization measured by SQUID magnetometry. Second, the conduction electrons of Au do not tend to have magnetic polarization induced by the magnetic impurities because the density of states at the Fermi level is small. It is well known that small amounts of magnetic impurities can affect the surrounding magnetic properties significantly and induce ferromagnetic moments for elements on the verge of ferromagnetism, such as Pd and Pt. However, to date, Au has not been reported to exhibit such an effect. Third, even if the magnetic impurity in the nanoparticles polarize nonmagnetic Au atoms, the amount of magnetic impurity is far smaller (< 5 ppm) than in Co/Cu multilayer systems, where a Cu spacer layer adjoins a Co magnetic layer of comparable volume. Thus, the signal in this case is extremely small and not detectable by XMCD. It should be noted that our system consists of Au and a polymer matrix that is made of nonmagnetic elements (C, H, N). Therefore, we conclude that the observed magnetization is an intrinsic effect of the Au particles.

The temperature dependence of ESM was also investigated and compared to the temperature variation of the SQUID magnetization. As shown in Fig. 3(b), ESM increases rapidly with decreasing temperature, although it seems that a finite constant value remains at high temperatures. The steep increase of ESM at low temperatures is consistent with the temperature variation of SQUID magnetization and does not contradict the characteristics of the superparamagnetic model [dotted line in Fig. 3(b)]. The temperature-independent magnetization corresponds to the Pauli-paramagnetic part mentioned above. This is not observed in bulk gold metal, which shows a diamagnetic response (-0.142×10^{-6} emu/g) and monotonic increase with decreasing temperature. We believe that the paramagnetism of Au is masked, in the bulk state, by the huge diamagnetism of the conduction electrons, but not in the case of nanoparticles, where paramagnetism emerges due to a size effect, reducing the density of states at

the Fermi level, and consequently suppressing the diamagnetism.

Our observation provides evidence for the existence of spin polarization in Au nanoparticles. Size effects, such as the so-called Kubo effect, cannot be responsible for the present results because the Kubo effect should be observed below 1 K for the size of nanoparticles studied here [3]. One possible explanation for the ferromagnetic polarization is surface ferromagnetism due to the large ratio of the number of atoms on the surface to the number in the core. It has been predicted theoretically that ferromagnetic spin polarization could take place in $4d$ and $5d$ transition metals with reduced coordination geometry [22]. Moreover, if one assumes that superparamagnetism and Pauli-paramagnetism arise from surface atoms and core atoms in Au nanoparticles, respectively, the observed mixture of superparamagnetism and Pauli-paramagnetism is reasonably explained.

On the other hand, the polymers, as stabilizers of the nanoparticles, also interact with the surface and would play an important role: the electronic properties of Au particles strongly depend on the type of polymers used [23]. For example, ligand molecules with a large affinity to the metal tend to quench the magnetic moment at the surface of nanoparticles [24]. Our recent experiments showed a sizeable reduction in the magnetization for ligand-stabilized nanoparticles such as dodecanthiol (DT)-stabilized nanoparticles. By contrast, several kinds of polymer-stabilized nanoparticles exhibited similar superparamagnetic behavior with large magnetic moments [25]. This suggests that the interaction between the surface of the Au nanoparticles and the ligand is weak for polymer-stabilized nanoparticles compared to DT-stabilized nanoparticles, in which it is known that the ligand forms covalent bonds with the metal surface [23]. This is related to the charge transfer process from the surface to the molecule [20,26]. Thus, the present experimental results suggest that PAAHC-Au can be regarded as a nearly freestanding cluster so that the surface magnetic moment survives. This is supported by recent experimental observations of ferromagnetism or paramagnetism in large bare Pd and Au clusters produced by a gas evaporation method [27,28]. For future prospects, the variation of the nanoparticle magnetism with the coupling strength between the protective agent and the surface is an interesting subject. The XMCD technique will be a powerful tool for this kind of study.

In summary, our XMCD and ESM experiments have revealed the intrinsic magnetic polarization in Au nanoparticles with a mean diameter of 1.9 nm. The external magnetic field and temperature dependence of the ESM signal suggested that magnetization of Au nanoparticles consist of a superparamagnetic part obeying the Curie law, and a temperature-independent Pauli-paramagnetic part. The mixture of these components is reasonably

explained by the picture that the surface atoms are ferromagnetic and the core atoms are Pauli-paramagnetic. We infer that the weak coupling between the protective agent and the gold surface is a prerequisite for the observation of spin polarization.

The synchrotron radiation experiments were performed at SPring-8 with the approval of Japan Synchrotron Radiation Research Institute (JASRI) as Nanotechnology Support Project of The Ministry of Education, Culture, Sports, Science, and Technology (Proposal No. 2002B0380-NS2-np/BL-No. 39XU).

*Present address: Graduate School of Pure and Applied Sciences, University of Tsukuba, 1-1-1 Tennodai, Tsukuba, Ibaraki 305-8571, Japan

- [1] S. Sun, C. B. Murray, D. Weller, L. Folks, and A. Moser, *Science* **297**, 1989 (2000).
- [2] R. P. Andres *et al.*, *Science* **273**, 1690 (1996).
- [3] Y. Volokitin *et al.*, *Nature (London)* **384**, 621 (1996).
- [4] *Cluster and Colloids: From Theory to Applications*, edited by G. Schmid (VCH, Weinheim, 1994).
- [5] *Physics and Chemistry of Metal Cluster Compounds*, edited by L. J. de Jongh (Kluwer, Dordrecht, 1994).
- [6] G. Schmid, *Polyhedron* **7**, 2321 (1988).
- [7] T. Teranishi, I. Kiyokawa, and M. Miyake, *Adv. Mater.* **10**, 596 (1998).
- [8] T. Teranishi and M. Miyake, *Chem. Mater.* **11**, 3414 (1999).
- [9] H. Hori *et al.*, *Phys. Lett. A* **263**, 406 (1999).
- [10] G. Schütz, M. Knulle, and H. Ebert, *Phys. Scr.* **T49A**, 302 (1993).
- [11] G. Schütz *et al.*, *Phys. Rev. Lett.* **58**, 737 (1987).
- [12] P. Gambardella *et al.*, *Nature (London)* **416**, 301 (2002).
- [13] D. Weller *et al.*, *Phys. Rev. Lett.* **75**, 3752 (1995).
- [14] M. Kucera *et al.*, *Phys. Rev. B* **66**, 144405 (2002).
- [15] M. Suzuki *et al.*, *Jpn. J. Appl. Phys.* **37**, L1488 (1998).
- [16] B. T. Thole, P. Carra, F. Sette, and G. van der Laan, *Phys. Rev. Lett.* **68**, 1943 (1992).
- [17] P. Carra, B. T. Thole, M. Altarelli, and X. Wang, *Phys. Rev. Lett.* **70**, 694 (1993).
- [18] J. Vogel *et al.*, *Phys. Rev. B* **55**, 3663 (1997).
- [19] F. Wilhelm *et al.*, *Phys. Rev. Lett.* **85**, 413 (2000).
- [20] P. Zhang and T. K. Sham, *Appl. Phys. Lett.* **81**, 736 (2002).
- [21] M. G. Samant *et al.*, *Phys. Rev. Lett.* **72**, 1112 (1994).
- [22] S. Blügel, *Phys. Rev. B* **51**, 2025 (1995).
- [23] P. Zhang and T. K. Sham, *Phys. Rev. Lett.* **90**, 245502 (2003).
- [24] D. A. van Leeuwen *et al.*, *Phys. Rev. Lett.* **73**, 1432 (1994).
- [25] H. Hori *et al.*, *Phys. Rev. B*, **69**, 174411 (2004).
- [26] H. Häkkinen, R. N. Barnett, and U. Landman, *Phys. Rev. Lett.* **82**, 3264 (1999).
- [27] T. Taniyama, E. Ohta, and T. Sato, *Europhys. Lett.* **38**, 195 (1997); T. Shinohara, T. Sato, and T. Taniyama, *Phys. Rev. Lett.* **91**, 197201 (2003).
- [28] W. H. Li *et al.*, *Phys. Rev. Lett.* **89**, 135504 (2002).